Atmospheric pressure chemical vapour deposition of fluorine-doped tin(IV) oxide from fluoroalkyltin precursors

Joanne E. Stanley¹ Anthony C. Swain¹, Kieran C. Molloy¹*, David W. H. Rankin², Heather E. Robertson² and Blair F. Johnston²

Received 6 May 2004; Revised 10 June 2004; Accepted 11 June 2004

Perfluoroalkytin compounds $R_{(4-n)}Sn(R_f)_n$ (R = Me, Et, Bu, $R_f = C_4F_9$, n = 1; R = Bu, $R_f = C_4F_9$, n = 2, 3; R = Bu, $R_f = C_6F_{13}$, n = 1) have been synthesized, characterized by 1H , ^{13}C , ^{19}F and ^{119}Sn NMR, and evaluated as precursors for the atmospheric pressure chemical vapour deposition of fluorine-doped SnO_2 thin films. All precursors were sufficiently volatile in the range $84-136\,^{\circ}C$ and glass substrate temperatures of $ca 550\,^{\circ}C$ to yield high-quality films with ca 0.79-2.02% fluorine incorporation, save for $Bu_3SnC_6F_{13}$, which incorporated <0.05% fluorine. Films were characterized by X-ray diffraction, scanning electron microscopy, thickness, haze, emissivity, and sheet resistance. The fastest growth rates and highest quality films were obtained from $Et_3SnC_4F_9$. An electron diffraction study of $Me_3SnC_4F_9$ revealed four conformations, of which only the two of lowest abundance showed close $F\cdot \cdot \cdot Sn$ contacts that could plausibly be associated with halogen transfer to tin, and in each case it was fluorine attached to either the γ - or δ -carbon atoms of the R_f chain. Copyright © 2005 John Wiley & Sons, Ltd.

KEYWORDS: fluoroalkyltin; fluorine-doped tin oxide; CVD; thin film

INTRODUCTION

Doped tin oxide films are arguably the most studied of the transparent conducting oxide thin films, as they have found widespread applications in various optoelectronic devices, including electrochromic displays, liquid crystal displays and solar cells.^{1–9} Of all the dopants employed (P,^{10,11}, As¹² Sb,^{13,14} etc.), fluorine is the most common as it generates films of high conductivity and optical clarity.^{15–17} Thin films of SnO₂, including fluorine-doped materials, can be achieved by a number of routes (sol–gel,^{18–23} RF sputtering,^{24,25} spray pyrolysis^{26–30}), though chemical vapour deposition (CVD) is the method of choice for large-scale coatings. This is particularly pertinent in the coating of architectural glass, where F: SnO₂ has been widely exploited as a solar control

*Correspondence to: Kieran C. Molloy, Department of Chemistry, University of Bath, Bath BA2 7AY, UK.

E-mail: k.c.molloy@bath.ac.uk Contract/grant sponsor: Engineering and Physical Sciences Research Council; Contract/grant number: GR/R17768.

Contract/grant sponsor: Pilkington.

device to reduce energy loss from buildings and in which the doped tin oxide film acts by being transparent to visible wavelengths but reflective in the infrared.^{31–33}

Conventional CVD experiments have employed both a volatile tin source [e.g. SnMe₄, $^{34-36}$ SnCl₄, $^{37-40}$ SnCl₂, 41 Sn(NMe₂)₄ 42] in conjunction with both oxygen (e.g. O_2 , 36 H₂O⁴³) and fluorine precursors (e.g. F_2 , 44 NH₄F, 45 HF, 16,46 BrCF₃, 15 various chlorofluorocarbons, 10 CF₃CO₂H^{29,47}). In comparison, relatively few attempts have been made to deposit fluorine-doped SnO₂ from a *single-source precursor*, though the advantages of eliminating toxic and/or environmentally unfriendly precursors from the CVD process are clear. Molecules that incorporate a direct Sn–F bond often lack sufficient volatility for CVD experiments by virtue of their tendency to incorporate Sn–F: \rightarrow Sn bridges (e.g. R₃SnF), 48,49 but species such as (β -diketonate)₂Sn(F)(OR) are now available and have been successfully employed in sol–gel approaches to F: SnO₂. 22,23,50

To retain volatility, most attempts to develop singlesource precursors for F:SnO₂ have incorporated the dopant

¹Department of Chemistry, University of Bath, Bath BA2 7AY, UK

²School of Chemistry, University of Edinburgh, West Mains Road, Edinburgh EH9 3JJ, UK

element into a more complex ligand system rather than having it directly bonded to tin, and successful CVD of $F:SnO_2$ has been achieved from $Sn[OCH(CF_3)_2]_4\cdot 2HNMe_2,^{51}$ $Sn[OCH(CF_3)_2]_2\cdot L$ ($L=HNMe_2$, $C_5H_5N),^{51,52}$ $Sn(O_2CCF_3)_2^{53}$ and $Bu_2Sn(O_2CCF_3)_2.^{54}$ To our knowledge, there is only one instance of the use of perfluoroalkytin species as precursors $[(C_4H_9C\equiv C)_3SnCH_2CH_2CF_3,\quad (C_4H_9C\equiv C)_3SnC_6H_4F-o]^{55,56}$ which have generated $F:SnO_2$ in sol–gel procedures by a γ -F transfer mechanism. Fully fluorinated alkyl chains may be anticipated to deliver the halogen to the tin from any of the available carbon centres, with a β -F transfer mechanism possibly the most likely (Scheme 1). 57 However, the possibility of alternative decomposition pathways, e.g involving radicals, cannot be excluded.

Fluoroorganotins have been known for many years, though they have proved less synthetically amenable than nonfluorinated organotins. The majority of known compounds contain only one fluorinated group (R₃SnR_f),⁵⁷⁻⁶⁰ with relatively few examples of compounds containing additional R_f groups, 61,62 particularly the homoleptic $Sn(R_f)_4$. 63 Of the known synthetic protocols (Table 1), reaction of hexaorganoditins with R_fI yields product mixtures that are often difficult to separate, 57-59 while fluoroorganomagnesium reagents are themselves relatively difficult to prepare.⁶¹ The latter can be more conveniently prepared indirectly, by reaction of RMgX with R_fI,⁶³ and this has enabled the Grignard route to afford a number of $(C_3F_7)_nSnR_{4-n}$ species in variable yields.63 Similar methodology has proved less effective in the case of organolithium reagents,60 though it has been used as a route to fluorovinyltin compounds.^{64,65} Less

$$\begin{array}{c} R \\ R \xrightarrow{--Sn} F \end{array} F \xrightarrow{-CF_2 = CF(R_f)} R_3SnF \xrightarrow{O_2 \\ 600^{\circ}C} SnO_2/F \end{array}$$

Scheme 1.

widespread methodologies have involved organocadmium reagents 66 and oxidative addition of $R_{\rm f}I$ to divalent tin species. 67

In this paper we report the synthesis of a range of $R_n Sn(R_f)_{4-n}$ and their use as atmospheric pressure CVD (APCVD) precursors for the deposition of $F:SnO_2$ thin films. Included in this report is the gas-phase structure of $Me_3(C_4F_9)Sn$, determined using a combination of electron diffraction and computational methods.

EXPERIMENTAL

General

Infrared spectra (cm⁻¹) were recorded as liquid films between NaCl plates using a Nicolet 510P FT-IR spectrophotometer, and elemental analyses were performed using an Carlo-Erba Strumentazione E.A. model 1106 microanalyser operating at 500 °C. 1H and 13C NMR spectra were recorded on a Jeol JNM-GX270 FT spectrometer and ¹⁹F and ¹¹⁹Sn NMR spectra were recorded on a Jeol JNM-EX400 FT machine, all using saturated CDCl₃ solutions unless indicated otherwise; chemical shifts are in parts per million with respect to either Me₄Si, Me₄Sn or CFCl₃, and coupling constants are in hertz. Details of our Mössbauer spectrometer and related procedures are given elsewhere;68 data are in millimetres per second. Dry solvents were obtained by distillation under inert atmosphere from the following drying agents: sodium-benzophenone (toluene, diethyl ether, tetrahydrofuran), calcium hydride (CH2Cl2), sodium (hexane). Standard Schlenk techniques were used throughout. Starting materials were obtained commercially and used without further purification.

Syntheses

*Tributyl(perfluorobutyl)tin, Bu*₃*SnC*₄*F*₉ (**1**) Isopropyl chloride (3.48 g, 44 mmol) in dry diethyl ether (30 ml) was added slowly to magnesium turnings (1.20 g,

Table 1. Known synthetic routes to fluoroalkyltin compounds

Compound Yield (%)		Synthesis ^a	Comments		
Me ₃ SnCF ₃	42			58	
$Me_3SnC_2F_5$	17	$R_3SnSnR_3 + R_fI$	Reaction involves Carius tube, difficult to	57	
$Me_3SnCF(CF_3)_2$	42		separate complex product mixture	57	
$Me_2Sn(C_2F_5)_2$	34			61	
$Bu_2Sn(C_2F_5)_2$	16	$Mg + R_fI$; $nR_fMgI + R_{4-n}SnX_n$	Slow reaction, low yields	61	
$Bu_3SnC_2F_5$	48		·	61	
$Sn(C_3F_7)_4$	19			63	
$MeSn(C_3F_7)_3$	11	i PrMgI + R _f I; n R _f MgI + R _{4-n} SnX _{n}	Good method, provided careful control of low	63	
$Me_2Sn(C_3F_7)_2$	82		temperature is maintained	63	
$Bu_3SnC_4F_9$	28	$MeLi + R_fI; nR_fLi + R_{4-n}SnX_n$	Method unreliable, complex product mixture	60	



50 mmol) to prepare the Grignard reagent isopropylmagnesium chloride. This was transferred by canula into a pressure-equalizing dropping funnel then added dropwise to freshly distilled perfluorobutyl iodide (15.32 g, 44 mmol) in dry diethyl ether (100 ml) at -78 °C. The solution was stirred at this temperature for 1 h to allow exchange to take place. Tributyltin chloride (9.72 g, 30 mmol) was then added slowly by syringe, the solution warmed to $-40\,^{\circ}\text{C}$ and stirred at this temperature for 2 h. The flask was then allowed to warm slowly to room temperature overnight with stirring, and the solvent removed in vacuo to yield an oil and a white solid. The mixture was extracted into $40^{\circ}-60^{\circ}$ petroleum ether and the solid removed by filtration. The solvent was removed in vacuo to leave a colourless oil, which by 119Sn NMR was found to be a mixture of tributylperfluorobutyltin and unreacted tributyltin chloride. The mixture was separated by column chromatography using silica gel as the stationary phase and 40° -60° petroleum ether as the eluant. Tributyl(perfluorobutyl)tin was eluted as the first fraction and removal of the solvent in vacuo yielded the product as a colourless oil (8.33 g, 55%). Anal. Found (calc. for C₁₆H₂₇F₉Sn): C, 37.8 (37.7)%; H, 5.37 (5.36)%. ¹H NMR: 0.92 [9H, t, $CH_3(CH_2)_3$], ${}^3J({}^1H-{}^1H)$ 7 Hz; 1.22 [6H, m, C₄H₉]; 1.35 [6H, m, C₄H₉]; 1.60 [6H, m, C₄H₉]. ¹³C NMR: 10.5 $[CH_3(CH_2)_3]; 13.5 [CH_3(CH_2)_2CH_2] [^1J(^{13}C-^{119}Sn) 333 Hz];$ 27.1 $[CH_3CH_2(CH_2)_2]$; 28.3 $[CH_3CH_2CH_2CH_2]$. C-F carbon atoms not observed. 19 F NMR: -126.6 [m, $CF_3CF_2(CF_2)_2$]; -119.2 [m, CF₃CF₂CF₂CF₂]; -118.2 [m, CF₃(CF₂)₂CF₂]; -81.7[m, $CF_3(CF_2)_3$]. ¹¹⁹Sn NMR: -1.6, t, ${}^2J({}^{119}Sn - {}^{19}F)$ 190 Hz. Mössbauer data: IS = 1.39; QS = 1.58. IR: 2961, 2928, 2859, 1466, 1420, 1379, 1346, 1235, 1154, 1084, 1007, 961, 793, 743.

Bis-(perfluorobutyl)dibutyltin, $Bu_2Sn(C_4F_9)_2$ (2)

The method described previously for 1 was utilized with isopropyl chloride (54.98 g, 63 mmol) added to magnesium turnings (1.60 g, 66 mmol) in dry diethyl ether (30 ml). The Grignard reagent was added dropwise to freshly distilled perfluorobutyl iodide (22.68 g, 66 mmol) at -78 °C. Dibutyltin dichloride (6.40 g, 21 mmol) dissolved in dry diethyl ether (5 ml) was added slowly by syringe at -78 °C, then the flask warmed to −40 °C and stirred at this temperature for 2 h. Following the previously described work-up procedure, bis-(perfluorobutyl) dibutyltin was obtained as a colourless liquid (4.58 g, 32%). Anal. Found (calc. for C₁₆H₁₈F₁₈Sn): C, 29.6 (28.6)%; H, 2.74 (2.71)%. ¹H NMR: 0.94 [6H, t, CH₃(CH₂)₃], $^{3}J(^{1}H-^{1}H)$ 7 Hz; 1.37 [4H, m, C₄H₉]; 1.60 [8H, m, C₄H₉]. ^{13}C NMR: $13.3 [CH_3(CH_2)_3]$; $13.6 [CH_3(CH_2)_2CH_2] [^1J(^{13}C-^{119}Sn)$ $358 \ Hz]; \ \ 26.9 \ \ [CH_3CH_2(CH_2)_2]; \ \ 27.4 \ \ [CH_3CH_2CH_2CH_2].$ C-F carbon atoms not observed. ¹⁹F NMR: -126.7 [m, $CF_3CF_2(CF_2)_2$; -118.4 [m, $CF_3CF_2CF_2CF_2$]; -113.9 [m, $CF_3(CF_2)_2CF_2$; -81.8 [m, $CF_3(CF_2)_3$]. ¹¹⁹Sn NMR: -56.0, quin, $^{2}J(^{119}Sn-^{19}F)$ 237 Hz. Mössbauer data: IS = 1.46; QS = 1.75. IR: 2965, 2932, 2865, 1468, 1348, 1238, 1132, 1071, 1009, 795, 745, 681, 644.

Tris-(perfluorobutyl)butyltin, $BuSn(C_4F_9)_3$ (3)

The methodology described for 1 was repeated with isopropyl chloride (6.44 g, 82 mmol) reacted with magnesium turnings (2.00 g, 82 mmol). The resultant Grignard reagent was added dropwise to freshly distilled perfluorobutyl iodide (29.19 g, 82 mmol) at $-78 \,^{\circ}\text{C}$. Butyltin trichloride (5.10 g,18 mmol) was added slowly by syringe and then the mixture warmed to -40 °C and stirred at this temperature for 2 h. Following the work-up procedure described for 1, 3 was obtained as a colourless liquid (2.74 g, 18%). Anal. Found (calc. for C₁₆H₉F₂₇Sn): C, 24.5 (23.1)%; H, 1.51 (1.09)%. ¹H NMR: 0.96 [3H, t, $CH_3(CH_2)_3$], ³ $J(^1H^{-1}H)$ 7 Hz; 1.42 [2H, m, C₄H₉]; 1.60 [4H, m, C₄H₉]. ¹³C NMR: 13.0 $[CH_3(CH_2)_3]; 13.2 [CH_3(CH_2)_2CH_2] [^1J(^{13}C-^{119}Sn) 395 Hz];$ 26.6 [CH₃CH₂(CH₂)₂]; 26.9 [CH₃CH₂CH₂CH₂]. C-F carbon atoms not observed. 19 F NMR: -126.8 [m, $CF_3CF_2(CF_2)_2$]; -117.8 [m, CF₃CF₂CF₂CF₂]; -108.6 [m, CF₃(CF₂)₂CF₂]; -81.9 $[m, CF_3(CF_2)_3]$. ¹¹⁹Sn NMR: -154.5, sept, ² $J(^{119}Sn^{-19}F)$ 300 Hz. Mössbauer data: IS = 1.44; QS = 1.49. IR: 2969, 2882, 1470, 1348, 1237, 1134, 1101, 997, 851, 777, 745, 683, 534.

Tributyl(perfluorohexyl)tin, Bu₃SnC₆F₁₃ (4)

The methodology described for 1 was employed with isopropyl chloride (1.89 g, 24 mmol) and magnesium turnings (0.60 g, 25 mmol). The resultant Grignard reagent was added to freshly distilled perfluorohexyl iodide (10.93 g, 25 mmol) at -78 °C and stirred at this temperature for 1 h. Tributyltin chloride (5.30 g, 16 mmol) was added slowly by syringe and then the flask was warmed to -40°C and stirred for a further 2 h. Following the work-up procedure described for 1, 4 was isolated as a colourless liquid (1.70 g, 17%). Anal. Found (calc. for C₁₈H₂₇F₁₃Sn): C, 35.5 (35.5)%; H, 4.45 (4.48)%. ¹H NMR: 0.92 [9H, t, $CH_3(CH_2)_3$], ³ $J(^1H^{-1}H)$ 7 Hz; $1.21\,[6H,m,C_4H_9];1.35\,[6H,m,C_4H_9];1.57\,[6H,m,C_4H_9].^{13}C$ NMR: $10.6 [CH_3(CH_2)_3]$; $13.5 [CH_3(CH_2)_2CH_2] [^1J(^{13}C-^{119}Sn)$ 329 Hz]; 27.1 [CH₃CH₂(CH₂)₂]; 28.3 [CH₃CH₂CH₂CH₂]. C-F carbon atoms not observed. ¹⁹F NMR: -126.7 [m, $CF_3CF_2(CF_2)_4$; -123.5 [m, $CF_3CF_2CF_2(CF_2)_3$]; -122.5 [m, $CF_3(CF_2)_2CF_2(CF_2)_2$; -118.3 [m, $CF_3(CF_2)_3CF_2CF_2$]; -117.9 [m, $CF_3(CF_2)_4CF_2$]; -81.4 [m, $CF_3(CF_2)_5$]. ¹¹⁹Sn NMR: -0.6, t, ${}^{2}J({}^{119}Sn - {}^{19}F)$ 191 Hz. Mössbauer data: IS = 1.35; QS = 1.57. IR: 2961, 2926, 2857, 1659, 1466, 1360, 1238, 1206, 1144, 1115, 1084, 1017, 882, 735, 652.

$Triethyl(perfluorobutyl)tin, Et_3SnC_4F_9$ (5)

The methodology described for 1 was followed using isopropyl chloride (2.80 g, 36 mmol) and magnesium turnings (0.90 g, 37 mmol). The resultant Grignard reagent was added to freshly distilled perfluorobutyl iodide (12.18 g, 35 mmol) at -78 °C, then triethyltin chloride (5.70 g, 24 mmol) added after stirring for 1 h. The flask was warmed to −40 °C and stirred for a further 2 h at this temperature. The work-up procedure described for 1 yielded 5 as a colourless liquid (3.68 g, 37%). Anal. Found (calc. for $C_{10}H_{15}F_9Sn$): C, 28.4 (28.3)%; H, 3.56 (3.57)%. ¹H NMR: 1.19 [9H, t, CH₃CH₂], ³J(¹H-¹H)



7 Hz; 1.27 [6H, q, CH₃CH₂]. 13 C NMR: 10.1 [CH₃CH₂]; 10.3 [CH₃CH₂] [1 J(13 C- 119 Sn) = 331 Hz]. C-F carbon atoms not observed. 19 F NMR: -126.7 [m, CF₃CF₂(CF₂)₂]; -119.5 [m, CF₃CF₂CF₂CF₂]; -117.9 [m, CF₃(CF₂)₂CF₂]; -81.8 [m, CF₃(CF₂)₃]. 119 Sn NMR: 3.6, t, 2 J(119 Sn- 19 F) 190 Hz. Mössbauer data: IS = 1.36; QS = 1.65. IR: 2957, 2878, 1470, 1383, 1348, 1237, 1154, 1084, 1047, 1006, 961, 743, 679, 519.

Trimethyl(perfluorobutyl)tin, $Me_3SnC_4F_9$ (6)

This was prepared by the same methodology as 1; yield 42%. 1 H NMR: 0.43 [9H, s, CH₃], 2 J(1 H- 119 Sn) 58.6 Hz. 13 C NMR: -8.86[CH₃], 1 J(13 C- 119,117 Sn) = 362, 346 Hz, 109.4 (q, CF₃) 1 J(19 F- 13 C) = 74 Hz, 112.4 (t, CF₂) 1 J(19 F- 13 C) = 61 Hz, 116.2 (t, CF₂) 1 J(19 F- 13 C) = 67 Hz, 119.1 (t, CF₂) 1 J(19 F- 13 C) = 67 Hz. 19 F NMR: -126.5 [t, CF₃CF₂CF₂CF₂] 3 J(19 F- 19 F) = 22 Hz; -121.2 [t, CF₃(CF₂)₂CF₂] 3 J(19 F- 19 F) = 23 Hz, 2 J(119 Sn- 19 F) = 225 Hz; -120.2 [t, CF₃CF₂(CF₂)₂] 3 J(19 F- 19 F) = 19 Hz; -81.7 [tt, CF₃(CF₂)₃] 3 J(19 F- 19 F) = 19 Hz, 4 J(19 F- 19 F) = 7 Hz. 119 Sn NMR: 25.3, t, 2 J(119 Sn- 19 F) 225 Hz.

Chemical vapour deposition

Details of our apparatus are given elsewhere.⁶⁹ The entire system consists of a horizontal cold-wall reactor with associated gas lines and electrical heater controls. The precursor was heated in a stainless-steel bubbler that was encased in an oven in which the temperature of the precursor could be measured accurately by a thermocouple positioned inside the bubbler. The pipework inside the oven contained a by-pass system that enabled the gas flows and temperatures to be set before the nitrogen carrier gas flow was turned to the bubbler to transport the vaporized precursor. Following the turning of the valves to direct the gas flow to the bubbler, the precursor was swept from the bubbler and then mixed with nitrogen diluent and oxygen before being transported from the oven. The mixture was then transported along the heated external pipework to the CVD reactor. Before the vapour reached the CVD reactor, it was passed through a baffle to promote laminar flow. After passing through the baffle, the precursor vapour was passed directly into the reactor chamber, which is 8 mm high, 40 mm wide and 300 mm long contained within ceiling tiles and walls of silica plates.

The glass substrate is positioned upon a large graphite susceptor, which is heated by three Watlow-fire rod cartridge heaters; the temperature of the graphite block is maintained by a Watlow series 965 controller, which monitors the temperature by means of thermocouples positioned inside the block. The graphite susceptor was held inside a large silica tube (330 mm long, 100 mm diameter) suspended between stainless-steel flanges upon which many of the electrical and gas line fittings are fixed. Air-tight seals are provided by 'Viton' O-rings.

All of the glass substrates were cleaned in an identical manner prior to use by washing thoroughly in sequence with tap water, copious amounts of distilled water and finally a generous amount of isopropyl alcohol, then allowed to drain. The glass was always prepared immediately prior to a deposition experiment to ensure as clean a substrate surface as possible.

The bubbler temperatures required to generate sufficient volatility for each precursor were 136 (1), 121 (2), 109 (3), 131 (4) and 84 °C (5). As all precursors were liquids, excessive temperatures for the heater tapes were not necessary as there were no problems with condensation of precursors within the pipework. Therefore, a consistent temperature of 200 °C was found to be satisfactory for the transport of all precursors between bubbler and reactor. Suitable gas flows were also found to be fairly universal for all compounds tested and those initially optimized for precursor 1 (diluent N_2 : 1.01 min⁻¹; carrier N_2 : 1.01 min⁻¹; O_2 : 0.61 min⁻¹) were found to be adequate for the other materials. Substrate temperatures were either 564 °C (1–4) or 546 °C (5) and the durations of the deposition processes were 15 (1), 7 (2), 10 (3), 25 (4) and 1 min (5).

Film analysis

The X-ray diffraction (XRD) equipment consisted of a Philips PW1130 generator operating at 45 kV and 40 mA to power a copper long fine-focus X-ray tube. A PW 1820 goniometer fitted with glancing-angle optics and proportional X-ray detector was used. The non-focusing thin-film optics employed a 0.25° primary beam slit to irradiate the specimen at a fixed incident angle of 1.5°. Diffraction radiation from the sample was collimated with a flat-plate collimator and passed through a graphite flat crystal monochromator to isolate diffracted copper $K\alpha$ peaks onto the detector. The equipment was situated in a total enclosure to provide radiation safety for the highly collimated narrow beams of X-rays. Data were acquired by a PW1710 microprocessor and processed using Philips APD VMS software. Crystalline phases were identified from the International Centre for Diffraction Data database. Samples of coating for XRD were of approximate dimensions $1.5 \text{ cm} \times 2.0 \text{ cm}$. Crystallite size was determined from line broadening using the Scherrer equation.⁷⁰ The instrumental effect was removed using the NIST SRM660 lanthanum hexaboride standard. These operating conditions were used in preference to conventional Bragg-Brentano optics for thin films to give an order of magnitude increase in count rate from a fixed volume of coating with little contribution from the substrate.

Film thickness was determined by etching a thin strip of the film with zinc powder and 50% HCl solution. This created a step in the film, which was measured with a Dektak stylus technique.

Haze was measured on a Pacific Scientific Hazeguard meter and with a barium fluoride detector. The calculation of haze was carried out by measurement of the specular light and diffusive light. Specular light is defined as light transmitted straight through the sample within $\pm 2.5^{\circ}$ of normal incidence and the diffusive light is defined as light scattered beyond 2.5° . The initial measurement was carried out with the specular detector slot closed and, therefore, a value for the sum of the

AOC

specular light and the diffusive light is obtained. The specular light slot is then opened and a measurement of the diffusive light is obtained.

Haze (%) = Diffusive light + Specular light/Diffusive light \times 100

Emissivity data were then calculated from the infrared reflectance spectra, measured using a two-beam Perkin Elmer 883 machine and measured against a rhodium mirror standard, using the formula:⁷¹

Emissivity =
$$1 - \frac{\int_{5 \mu m}^{50 \mu m} R_{\lambda} P_{\lambda} d\lambda}{\int_{5 \mu m}^{50 \mu m} P_{\lambda} d\lambda}$$

i.e. integral of total emittance between 5 and 50 μ m divided by the integral from 5 to 50 μ m of the total emittance of a blackbody at room temperature.

Sheet resistance was measured with a four-point probe on an electrically isolated scribed circle of film ($\square = 25 \text{ cm}^2$) and corrected using a conversion factor, the value being dependent on the diameter of the scribed circle. Knowing

the sheet resistance and the film thickness, the resistivity was determined by:

Resistivity (Ω cm) = Sheet resistance (Ω/\square) × Thickness (cm)

X-ray fluorescence measurements were made on a Philips PW1400 machine fitted with a scandium target X-ray tube. The penetration depth achieved was between 9 and 10 μm , so the result obtained was throughout the thickness of the coating. The analysis was performed on approximately 6 cm² of material.

Theoretical methods

Calculations were performed on a DEC Alpha APX 1000A workstation using the GAUSSIAN 94 program. An extensive search of the potential energy surface of $Me_3SnC_4F_9$ (6) was undertaken at the $HF/3-21G^*$ level in order to locate all structurally stable conformers. In total four conformers were found, all with C_1 symmetry (Fig. 1). Further geometry optimisations were then undertaken for all minima with the D95 basis set (a full double zeta basis set) including a double zeta set on tin (Dunning TH, unpublished results; 15s, 11p, 7d/11s, 7p, 4d) at the HF level. The subsequent two sets of calculations used the LanL2D Z^{74-76} effective core potential

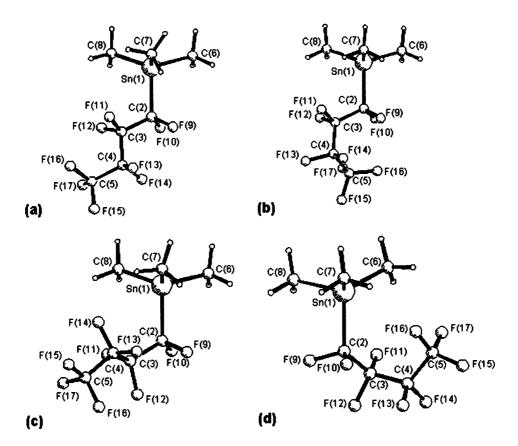


Figure 1. Ab initio structures and numbering schemes of the conformers of Me₃SnC₄F₉: (a) lowest energy conformer (1); (b) conformer 2, 0.8 kJ mol⁻¹ above the minimum; (c) conformer 3, 4.5 kJ mol⁻¹ higher in energy than the minimum; (d) conformer 4, 16.6 kJ mol⁻¹ above the minimum.

basis set (incorporating relativistic effects) for tin and D95 for the remaining atoms at the HF and MP2 levels of theory. Vibrational frequencies were calculated from analytic second derivatives up the D95 (fluorine, carbon, hydrogen, oxygen), LanL2DZ (tin)/HF level to confirm all conformers as local minima on the potential energy surface. The force constants obtained from these calculations were subsequently used to construct harmonic force fields for all conformers using the ASYM40 program.⁷⁷ As no fully assigned vibrational spectra are available for ${\bf 6}$ to scale the force fields, a scaling factor of 0.9was adopted for bond stretches, angle bends and torsions.⁷⁸ The scaled harmonic force fields were then used to provide estimates of amplitudes of vibration u for use in the gas-phase electron diffraction (GED) refinements. The results of the highest level of these theoretical calculations [MP2/LanL2DZ (tin) -D95] can be seen in Table 2; a complete tabulation of the results at all levels is available as supplementary data.

Gas-phase electron diffraction

Data were collected on Kodak Electron Image plates using the Edinburgh gas diffraction apparatus.⁷⁹ An accelerating voltage of ca 40 kV (electron wavelength ca 6.0 pm) was used, with sample and nozzle temperatures of ~400 K and \sim 450 K respectively; nozzle-to-plate distances were 95.57 and 252.33 mm. The weighting points for the off-diagonal weight matrices, correlation parameters and scale factors for the two camera distances are given in Table 3, together with electron wavelengths. The wavelengths were determined from the scattering patterns of benzene vapour, recorded immediately after the patterns of Me₃SnC₄F₉ and analysed in exactly the same way, to minimize systematic errors in wavelengths and camera distances. A PDS densitometer at the Institute of Astronomy in Cambridge was used to convert the intensity patterns into digital form. Data reduction and least-squares refinements were carried out using the new 'ed@ed' program (Johnston BF, Rankin DWH, Turner AR)80, employing the scattering factors of Ross et al.81

Based on the ab initio molecular orbital (MO) calculations, a theoretical model was written for Me₃SnC₄F₉, allowing for the co-existence of three conformers, each with C_1 symmetry. The three lowest energy conformers found in the ab initio study (Fig. 1a-c) were included in the model. The fourth conformer, which was 16.6 kJ mol⁻¹ higher in energy relative to the lowest energy conformer, was ignored, as it would only have made up a tiny percentage of the conformer mixture at the experimental temperature. Twenty-eight parameters were required in order to model the compound in the desired symmetry. These consisted of eight bonded-distance parameters (four average bond lengths and four difference parameters), 18 angle parameters, and two parameters that controlled the relative amounts of conformers. In the description of the parameters given below, the term 'average' applies to the average for all conformers modelled. The distance parameters are the average C-H (p_1) , the average Sn-C (p_2) , the difference between the average of Sn(1)-C(6), Sn(1)-C(7) and

Table 2. Calculated geometrical parameters (distances/pm, angles/deg) for the three lowest energy conformers of $Me_3SnC_4F_9$ calculated at the MP2/LanL2DZ (Sn) - D95 level

	Conformer 1	Conformer 2	Conformer 3
Bond lengths			
Sn(1)-C(6)	213.3	213.2	213.2
Sn(1) - C(0) Sn(1) - C(7)	213.3	213.2	213.2
Sn(1) - C(7) Sn(1) - C(8)	213.3	213.1	213.2
Sn(1) - C(3) Sn(1) - C(2)	213.5	213.3	213.2
C(2)-C(3)	152.4	152.2	152.7
C(3)-C(4)	153.9	153.9	153.5
C(4)-C(5)	153.7	153.4	153.6
C(2)-F(9)	138.9	138.5	138.5
C(2)-F(3) C(2)-F(10)	138.1	138.6	138.9
C(3)-F(11)	137.0	137.2	136.3
C(3)-F(11) C(3)-F(12)	136.6	136.3	136.5
	135.6	135.6	135.7
C(4)-F(13)			136.9
C(4)-F(14)	135.9	136.2	134.4
C(5)–F(15) C(5)–F(16)	134.2 134.4	134.4 134.1	134.4
	134.4	134.1	134.2
C(5)-F(17)	134.0	134.2	134.2
Inter-bond angles	1000	100 =	1010
C(2)-Sn(1)-C(6)	103.2	103.7	101.8
C(2)-Sn(1)-C(7)	105.8	105.7	107.3
C(2)-Sn(1)-C(8)	106.9	107.5	107.6
C(7)-Sn(1)-C(8)	113.6	112.9	113.9
Sn(1)-C(2)-C(3)	115.3	115.3	121.7
C(2)-C(3)-C(4)	116.3	119.2	114.7
C(3)-C(4)-C(5)	114.5	117.2	115.4
F(9)-C(2)-F(10)	107.1	107.1	104.2
F(11)-C(3)-F(12)	107.9	108.1	108.6
F(13)-C(4)-F(14)	109.3	108.8	108.4
Sn(1)-C(2)-F(9)	109.5	109.9	109.7
Sn(1)-C(2)-F(10)	110.7	110.5	107.3
C(2)-C(3)-F(11)	107.9	107.6	110.2
C(2)-C(3)-F(12)	108.1	108.0	108.0
C(3)-C(4)-F(13)	108.7	109.0	110.2
C(3)-C(4)-F(14)	109.5	106.9	107.7
C(4)-C(5)-F(15)	109.2	108.8	108.9
C(4)-C(5)-F(16)	110.1	110.6	110.5
C(4)-C(5)-F(17)	110.8	110.5	110.4
Torsion angles			
C(3)-C(2)-Sn(1)-C(6)	166.2	157.3	164.6
Sn(1)-C(2)-C(3)-C(4)	164.8	167.1	55.7
C(2)-C(3)-C(4)-C(5)	161.6	53.5	168.4
C(3)-C(4)-C(5)-F(15)	169.1	169.5	169.3
C(2)-Sn(1)-C(6)-H(21)	179.8	178.5	178.6
H(20)-C(7)-Sn(1)-C(6)	31.3	62.8	50.4
H(26)-C(8)-Sn(1)-C(6)	63.4	173.2	53.6

Table 3. Nozzle-to-plate distances, weighting functions, correlation parameters, scale factors and electron wavelengths used in the electron-diffraction study of Me₃SnC₄F₉

Nozzle-to-plate distance (mm) ^a	95.57	252.33
$\Delta s (\mathrm{nm}^{-1})^{-}$	40	20
$s_{\min} (\text{nm}^{-1})$	112	30
$s_{\rm w1} \ ({\rm nm}^{-1})$	132	50
$s_{\rm w2} \ ({\rm nm}^{-1})$	212	94
$s_{\text{max}} (\text{nm}^{-1})$	248	110
Correlation parameter	0.0185	-0.3792
Scale factor ^b	0.343(16)	0.895(10)
Electron wavelength (pm)	6.016	6.016

^a Determined by reference to the scattering pattern of benzene vapour.

Sn(1)–C(8) and the longer Sn(1)–C(2) (p_3), the average C–C (p_4) , the average C-F (p_5) , the difference between the average of C(2)-F(9/10) and the average of the other C-F bonds (p_6) , the difference between the average of C(3)-F(11/12) and the average of the C(4)-F(13/14) and C(5)-F(15/16/17)bonds (p_7) and the difference between the average of C(4)-F(13/14) and the average C(5)-F(15/16/17) (p_8). The angle parameters are the average $C(4)-C(5)-F_{(terminal)}$ angle (assuming local C_3 symmetry on the terminal CF_3 group) (p_9), the average F-C-F angle for the three CF_2 groups (p_{10}) , C(3)-C(4)-C(5) (in conformer 2) (p_{11}) , C(2)-C(3)-C(4) (in conformer 2) (p_{12}) , Sn(1)–C(2)–C(3) (in conformer 2) (p_{13}) , C(2)-Sn(1)-C(6) (p_{14}), C(7)-Sn(1)-C(8) (p_{15}), the average of C(7&8)-Sn(1)-C(6) (assuming local C_s symmetry of the SnMe₃ group, with C(2)-Sn(1)-C(6) lying on the mirror plane) (p_{16}) and the average Sn–C–H angle (p_{17}). Parameters p_{18} , p_{19} and p_{20} have the same definitions as parameters p_{11} , p_{12} and p_{13} but apply to conformer 1. Parameters p_{21} , p_{22} and p_{23} also have the same definitions as parameters p_{11} , p_{12} and p_{13} but apply to conformer 3. Parameter p_{24} is a torsion angle used for the approximately anti-configured C-C-C-C torsion angles in all three conformers. These include C(3)-C(2)-Sn(1)-C(6), Sn(1)-C(2)-C(3)-C(4) and C(2)-C(3)-C(4)-C(5) in conformer 1, C(3)-C(2)-Sn(1)-C(6)and Sn(1)-C(2)-C(3)-C(4) in conformer 2 and C(3)-C(2)-Sn(1)-C(6) and C(2)-C(3)-C(4)-C(5) in conformer 3. While individual torsion angles could have been used as unique independent parameters, it was felt that the calculated values were sufficiently similar to warrant grouping them together in this way. Fixed difference values were assigned to parameter p_{24} , with values of -3.0° for C(2)-C(3)-C(4)-C(5) in conformer 1 ($p_{24} - 3.0^{\circ}$) and 6.0° for C(2)–C(3)–C(4)–C(5) in conformer 3 ($p_{24} + 6.0^{\circ}$). A parameter, p_{25} , was also needed to model the approximately gauche C-C-C-C torsion angles, C(2)-C(3)-C(4)-C(5) in conformer 2 and Sn(1)-C(2)-C(3)-C(4) in conformer 3. The torsion angle C(3)-C(4)-C(5)-F(15), which controls the rotation of the terminal CF_3 group in all conformers, is p_{26} . The final two parameters, p_{27} and p_{28} , control the fractional amounts of conformers 2 and 1 respectively, with the weight of conformer 3 automatically being calculated as $1.0 - (p_{27} + p_{28})$. The refined and calculated parameters can be found in Table 4.

The starting parameters for the r_a refinements were taken from the theoretical geometry optimized at the MP2 level. The theoretical Cartesian force field was obtained and converted into a force field described by a set of symmetry coordinates using a version of the ASYM40 program modified to work for molecules with more than 40 atoms. All 28 geometric parameters and 26 groups of vibrational amplitudes were refined, using 24 flexible restraints on geometric parameters (Table 4) and 25 restraints on amplitudes of vibration (see supplementary data) according to the SARACEN method. 82,83

RESULTS AND DISCUSSION

Synthesis and spectroscopy

A series of perfluoroalkytin compounds $R_{(4-n)}Sn(R_f)_n$ (1-6) has been prepared employing the methodology of Seyferth et al.:63

$$\begin{split} Mg + {}^{\mathrm{i}}PrCl &\xrightarrow{Et_2O} {}^{\mathrm{i}}PrMgCl \xrightarrow{R_fI/Et_2O} R_fMgCl + {}^{\mathrm{i}}PrI \\ nR_fMgCl + R_{(4-n)}SnCl_n &\xrightarrow{Et_2O} R_{(4-n)}Sn(R_f)_n + nMgCl_2 \\ R = Bu, R_f = C_4F_9, n = 1 \ (1), n = 2 \ (2), n = 3 \ (3) \\ R = Bu, R_f = C_6F_{13}, n = 1 \ (4) \\ R = Et, R_f = C_4F_9, n = 1 \ (5) \\ R = Me, R_f = C_4F_9, n = 1 \ (6) \end{split}$$

The synthesis of $Bu_3SnC_4F_9$ (1) illustrates the importance of choice of synthetic protocol in realizing viable yields of the fluoroalkyltin compounds. Direct synthesis of the Grignard C₄F₉MgI from magnesium and C₄F₉I followed by addition of Bu₃SnCl yielded a crude product mixture (¹¹⁹Sn NMR) consisting of unreacted organotin (155.9 ppm) in addition to a quantity of Bu_3SnI (86.2 ppm) and $Bu_3SnC_4F_9$ (-1.6 ppm). Vacuum distillation of the mix was unsuccessful in yielding a clean product due to the similarity in boiling points of all the tributyltin compounds. Separation by column chromatography required a very long column (~30 cm) as the triorganotins eluted only slightly more slowly from the column than the desired tetraorganotin. The perfluoroalkyltin compound was ultimately isolated in a very poor yield (10%).

Similarly, the organolithium reagent C₄F₉Li, prepared in situ from MeLi and C₄F₉I, required an extremely low temperature $(-100 \,^{\circ}\text{C})$ to prevent decomposition, resulting in poor conversion to the desired product (1; 11% after column chromatography, as above). These findings are consistent with a previous report on this reaction.⁶⁰

The most effective and reproducible method for the synthesis of the fluoroalkyltins involved the indirect

^b Values in parentheses are the estimated standard deviations.



Table 4. Refined and calculated geometric parameters for Me₃SnC₄F₉ (distances in picometres, angles in degrees) from the GED study^{a,b}

No.	Parameter	GED (r_a)	MP2/Lan2DZ (Sn) $-D95$	Restraint
$\overline{p_1}$	C–H av.	109.5(7)	109.3	109.8(13)
p_2	Sn-C av.	214.7(5)	215.4	215.6(23)
p_3	[Sn(1)-C(2)] - [Sn(1)-C(6/7/8)]	9.0(5)	8.4	9.0(5)
p_4	C–C av.	155.0(3)	153.2	154.8(4)
p_5	C–F av.	135.9(1)	136.1	136.0(32)
p_6^c	$\Delta C(2) - F(9/10)$	3.3(2)	3.2	3.3(2)
p_7^c	$\Delta C(3) - F(11/12)$	1.9(1)	1.7	1.9(1)
p_8^c	$\Delta C(4) - F(13/14)$	1.3(1)	1.8	1.3(1)
p_9	C(4)-C(5)-F(15/16/17) av.	109.7(4)	109.9	110.2(5)
p_{10}	F-C-F	107.7(4)	107.7	107.7(5)
p_{11}	C(3)-C(4)-C(5) in conf. 2	117.0(8)	117.2	117.2(10)
p_{12}	C(2)-C(3)-C(4) in conf. 2	120.0(5)	119.2	120.2(5)
p_{13}	Sn(1)-C(2)-C(3) in conf. 2	116.8(9)	115.3	115.7(11)
p_{14}	C(2)-Sn(1)-C(6)	102.9(5)	102.9	102.8(5)
p_{15}	C(7)-Sn(1)-C(8)	107.0(9)	113.5	106.5(10)
p_{16}	C(6)-Sn(1)-C(7/8) av.	113.4(5)	112.9	113.4(5)
p_{17}	Sn-C-H av.	110.7(5)	110.8	110.8(5)
p_{18}	C(3)-C(4)-C(5) in conf. 1	114.4(5)	114.5	114.5(5)
p_{19}	C(2)-C(3)-C(4) in conf. 1	116.7(5)	116.3	117.0(5)
p_{20}	Sn(1)-C(2)-C(3) in conf. 1	117.6(7)	115.3	115.5(13)
p_{21}	C(3)-C(4)-C(5) in conf. 3	115.9(5)	115.4	115.9(5)
p_{22}	C(2)-C(3)-C(4) in conf. 3	114.2(10)	114.7	114.1(10)
p_{23}	Sn(1)-C(2)-C(3) in conf. 3	122.0(14)	121.7	122.0(15)
p_{24}^{c}	Anti twist	164.6(17)	164.3	
p_{25}^{c}	Gauche twist	12.1(23)	54.6	
p_{26}	CF ₃ twist	184.0(19)	169.3	169.2(5)
p_{27}	Weight conf. 2	0.65	0.49	
p_{28}	Weight conf. 1	0.33	0.38	

^a Figures in parentheses are the estimated standard deviations of the last digits.

formation of the Grignard, R_fMgI, from ⁱPrMgCl and C₄F₉I. Although this reaction was also found to be extremely temperature sensitive and care had to be taken to ensure the temperature remained at -78 °C during this step, adequate formation of the perfluoromagnesium iodide could be realized over a minimum of 1 h. Reagent quantities were chosen to form 50% excess equivalents of the required Grignard reagent (based on 100% conversion) for reaction with the organotin chloride to allow for decomposition and incomplete conversion. This was especially important for the synthesis of compounds in which more than one chloride was to be replaced, to reduce the complexity of the final product mixture. The reaction of the fluorinated Grignard reagent with the organotin chloride proceeded better at a slightly elevated temperature (-40°C), and to achieve a reasonable yield this temperature had to be maintained for at least 2 h. The yield of 1 by this method was increased to 55%.

It was generally found that the yields decreased as more fluorinated groups were introduced, and overall were in the range 18-55%. All compounds are air-stable, colourless liquids.

NMR spectra confirm the composition of the compounds. The magnitudes of the $^1J(\text{Sn-C}_H)$ for 1-6 (331–395 Hz) are in the appropriate region for four-coordinate organotin species (300–400 Hz). $^{84.85}$ There is an increase in $^1J(\text{Sn-C}_H)$ as the number of R_f groups incorporated increases (1: 333 Hz; 2: 358 Hz; 3: 395 Hz), due to the high electronegativity of the R_f ligands, which have a stronger demand for 5p(Sn) character in the Sn–C $_F$ bond as the R groups are progressively replaced by R_f ligands; the s-character of the Sn–C $_H$ bonds increases accordingly.

¹¹⁹Sn NMR proved the best measure of compound purity and validation of the number of R_f groups introduced. All the compounds showed strong coupling of tin to the fluorine atoms of the α -carbon, but there were no couplings

^b Unless stated otherwise, parameter definitions apply to all three conformers.

^c See text for a full definition.

with more distant fluorine atoms displayed. The 2n + 1multiplicity (observation of the triplet, quintet or septet) of the tin signal proved diagnostic of the presence of two, four and six α -fluorine atoms respectively. The upfield shift of $\delta(^{119}\text{Sn})$ in the series 1-3 (-1.6 ppm, -56.0 ppm, -154.5 ppm respectively) is consistent with other reports (e.g. $Me_2Sn(C_3F_7)_2$, -22.8 ppm; $MeSn(C_3F_7)_3$, -131 ppm). $^{2}I(^{119}Sn-^{19}F)$ increases as more $R_{\rm f}$ groups are introduced (e.g. 1: 190 Hz; 2: 237 Hz; 3: 300 Hz), which is also consistent with earlier reports (e.g. $MeSn(C_3F_7)_3$, 308 Hz; $Sn(C_3F_7)_4$, 387 Hz).63

The electronegativity imbalance between R and R_f is also manifest in observable Mössbauer quadrupole splittings for 1-5 (1.49–1.75 mm s^{-1}). The QS for Me₃SnCF₃ has been reported as $1.38 \text{ mm s}^{-1.86}$

Structural study

The computational study of Me₃SnC₄F₉ (6) showed that there are four stable conformers (Fig. 1a-d), two of which are likely to be significantly populated at the temperature of the electron diffraction experiments (ca 450 K). This is supported by the analysis of the GED data, which shows that the only conformers in which a fluorine atom gets close enough to tin for exchange to occur easily (conformers 3 and 4) must only exist in small quantities (ca 2% for conformer 3, <1% for conformer 4). However, the experimental data indicate that even a large fraction of conformer 3, in which the C₄F₉ chain is less twisted than in the computed high-energy conformer 4, has a negative effect on the refinement.

The optimum fit of the GED data was found for a mixture of 65% conformer 2, 33% conformer 1 and 2% conformer 3, whereas calculations predict that the mixture is 86% conformer 2 and 13% conformer 1 (1% conformer 3). These values were obtained by manual adjustment of the two weight parameters, p_{27} and p_{28} .

The geometrical parameters obtained in the refinement are listed in Table 4; interatomic distances and amplitudes of vibration, and lists of the most significant elements of the least-squares correlation matrix are available as supplementary data. The final R factors were $R_G = 0.027$ and $R_D = 0.039$. Figures showing the experimental and final difference molecular scattering intensity curves and the radial distribution curves are available as supplementary data. Elucidation of the GED structure of Me₃SnC₄F₉ was made particularly tricky by the poor quality of the data recorded at the short camera distance, although the experiment was repeated several times. This is reflected in the large number of restraints that were employed in the refinement. Without these restraints even parameters such as Sn-C and C-F, which would normally be expected to refine well, were not particularly well behaved and were found to refine to unrealistic values. It is unlikely that a successful refinement could have been carried out in this instance without the results from the ab initio calculations. Comparing the Sn-C bond lengths from the GED experiment with those calculated ab initio at the MP2 level of theory, it was found that the

average was 0.7 pm shorter in the GED refinement (214.7 pm GED, 215.4 pm MP2). The calculations that did not use the LanL2 pseudo-potential on tin predicted the average Sn-C bond length to be 217.9 pm. Inclusion of the pseudo-potential results in a shortening of the Sn-C distance by 2.1 pm at the HF level and 2.3 pm at the MP2 level. The GED value suggests that using a pseudo-potential on tin is important in order to obtain accurate theoretical structures.

Calculations also found the Sn(1)-C(2) bond of the fluoroalkyltin moiety to be longer than the other Sn-C bonds by around 9 pm, reflecting the presence of the fluorine substituents on the butyl chain. This is consistent with the trend in ¹J(¹¹⁹Sn-¹³C_H) couplings, which indicates a concentration of 5p(Sn) character in the Sn-C_F bonds; hence the relative lengthening of the latter. A 9.0(5) pm difference was also found in the GED study, although this value was restrained quite closely to that calculated at the MP2 level. C-C bond lengths varied very little on changing the level of the theoretical calculations. A decrease of around 0.9 pm (from 154.2 pm to 153.3 pm) was observed on going from the HF/D95 to the MP2 level. There was reasonable agreement between these values and the value of 155.0(3) pm from the GED refinement. The average C-F bond length was found to vary more than the C-C distances. At the MP2 level the average C-F distance was found to be 136.1 pm, around 3 pm longer than in both the HF/D95 and HF/LanL2DZ(Sn)-D95(F,C,H) calculations. This is likely to be a result of better treatment of the lone pairs on the fluorine atoms with the inclusion of electron correlation at the MP2 level. The GED value of 135.9(1) pm for the average C-F bond length is in excellent agreement with that obtained in the MP2 calculations. Similarity between the values calculated at the MP2 and HF/3-21G* levels (both 136.1 pm) is fortuitous, a consequence of cancellation of errors in the HF/3-21G* calculations.

Bond angles generally varied little in the series of ab initio calculations. However, there was a substantial difference in the angle Sn(1)-C(2)-C(3) for the three major conformers, being calculated as 115.3° for both conformer 1 and conformer 2, but as much as 121.7° for conformer 3. Despite the fairly large differences between conformers, <Sn(1)-C(2)-C(3) did not change much with different theoretical treatments, decreasing by 1.3 to 1.5° from the HF/D95 level to the MP2 level in each case. C(2)-C(3)-C(4) also varied with conformation, but the extent of this variation changed little with the level of calculation. The GED values for these angles in the three conformers are in good agreement with theory, although once again restraints were necessary to obtain stability in the refinement. C(3)-C(4)-C(5) varied little with the level of calculations, and differences between conformers were smaller, with values of 114.5° in conformer 1, 117.2° in conformer 2 and 115.4° in conformer 3 at the MP2 level. The bond angles with tin at the apex [C(2)-Sn(1)-C(6)], C(7)-Sn(1)-C(8), C(2)-Sn(1)-C(7) and C(2)-Sn(1)-C(8)] did not show the same sensitivity to the inclusion of the pseudopotential on tin as the Sn-C bond lengths, and angles changed by less than 1° as the level of calculation was varied. The average F–C–F angle for the CF₂ groups also changed little, even on going to the MP2 level of theory, despite the C–F bond lengths being significantly longer at this level.

The unrestrained anti twist parameter, p_{24} , which controlled the torsion angle C(2)–C(3)–C(4)–C(5) in conformers 1 and 3, was found to agree well with the value calculated *ab initio*, the GED refinement giving $164.6(17)^{\circ}$, compared with the calculated (MP2) values of 161.6° in conformer 1 and 168.4° in conformer 3. The gauche twist parameter, p_{25} , which controls the torsion angles C(2)–C(3)–C(4)–C(5) in conformer 2 and Sn(1)–C(2)–C(3)–C(4) in conformer 3, was, however, found to be substantially smaller in the GED refinement.

Of significance to the CVD study is the relative disposition of tin and a suitable fluorine atom, such that transfer of the halogen to the metal can readily occur, thereby affording a doped SnO₂ film. In all of the four conformers the fluorines of the α -CF₂ group are within 300 pm of tin, but are oriented away from the metal. In addition, the three low-energy conformers (Fig. 1a-c) have close Sn···F contacts with the fluorine atoms attached to the β -carbon, with Sn···F(11) distances between 320 and 330 pm, although in none of these cases does the halogen approach the tin trans to a CH₃ group, a scenario which would be most effective in transferring the halogen to the tin via a five-coordinated transition state. Furthermore, there is no apparent lengthening of the C-F bond associated with the short Sn···F contact. In contrast, conformers 3 and 4 offer more realistic pathways for fluorine transfer to tin. The high-energy, low-abundance conformer 4 has a close contact with the fluorine atom of the δ -carbon [Sn···F(17); 322 pm] in which such a five-coordinated arrangement, albeit somewhat distorted [<F(17) \cdots Sn-C(8): 163.2°] is observed. In conformer 3, the closest $F \cdot \cdot \cdot Sn$ contact is 327.4 pm, with an F(14)···Sn−C(6) angle of 165.3°. In both conformer 3 and conformer 4 the C-F that is thus linked to the tin atom is calculated to be 1.2 pm longer than its neighbouring C-F bond(s).

Film deposition

Compounds 1–5 have been screened as potential CVD precursors for F:SnO₂, using an APCVD reactor described elsewhere.⁶⁹ In all cases, the substrate used was 4 mm glass that had been undercoated with a thin film of SiCO to act as a blocking layer to prevent sodium diffusion into the deposited film. The reactor temperatures (*ca* 550 °C) were found to produce good transparent films that adhered well to the glass substrate and could not be removed without relatively harsh treatment. Lower temperatures resulted in a vast decrease in the growth rates, whereas powdery deposits were obtained at higher temperatures. No attempt has been made in this study to optimize deposition parameters.

It was found that precursor (bubbler) temperatures generally in excess of $100\,^{\circ}\text{C}$ were required for reasonable growth rates, suggesting that the precursors are not remarkably volatile. The butyltin derivatives 1-4 were found to require much higher temperatures ($109-136\,^{\circ}\text{C}$) than the ethyltin compound 5 ($84\,^{\circ}\text{C}$), which suggests a higher degree

of volatility for precursors containing smaller R groups. Also, it was observed that the acceptable bubbler temperature decreased through the series 1-3 (136°C, 121°C, 109°C respectively) as the number of fluorinated groups increased, showing a greater volatility with additional fluorine incorporation. Consequently, deposition times required to produce films of thickness adequate for characterization (ca 300 nm) varied for $R_{4-n}Sn(R_f)_n$ in the order Bu > Et and $n = 1 > 2 \approx 3$ (see Experimental). Films grown from the butyltin compounds 1-4 were found to favour deposition at the front end of the substrate directly after the precursor inlet and only coated the first 5-6 cm of the glass, whereas the film grown from the ethyltin derivative 5 had a much more uniform appearance and spanned the total length of the glass. Therefore, for commercial production, where a high growth rate and consistent film uniformity are required, small R groups appear to be preferential. Et₃SnC₄F₉ (5), which required a bubbler temperature of 84 °C and deposited ca 300 nm thickness films of F:SnO2 in 1 min at 546 °C, was the most effective of the precursors evaluated in this study.

Film characterization

Glancing-angle XRD confirmed the film composition as crystalline SnO_2 in all cases; a representative diffraction pattern is shown in Fig. 2 for the film grown from 1. From line broadening measurements of the (110) reflection it was possible to estimate the approximate crystallite sizes of the samples, which lie in the range 11.9-26.4 nm (data to 3, 4 represent the two extremes respectively). All the films showed similar preferred orientations when compared with a standard sample of SnO₂. The degree of orientation can be quantified as the ratio of the intensity of the (200) reflection to the total integrated intensity of the diffraction pattern, expressed as a percentage; for a random orientation of SnO₂ this ratio would be 7%. The degree of preferred (200) orientation is at a minimum for films grown from the tributyltin derivatives 1 and 4 (13.8% and 9.5% respectively), increasing as the number of R_f groups increases (2: 17.8%; 3: 19.2%) but maximizing for the triethyltin precursor 5 (25.5%). Previous work has shown that SnO₂ films grown along the (200) direction contain less structural defects^{10,40} and, therefore, give better performance for such applications as solar cells.87

Scanning electron microscopy of the film deposited from 1 (Fig. 3) shows a uniform morphology and a smooth film surface; the photograph clearly shows the SiCO undercoat on which the fluorine-doped tin oxide coating has subsequently been deposited.

For all films, the thickness, haze, emissivity, sheet resistance, resistivity and fluorine content were measured (Table 5) and compared with data typical of $F:SnO_2$ films used in solar control coatings.⁸⁸ Fluorine incorporation in the range 1-8%, ^{15,89} optimizing at ca 3%, ⁹⁰ has been found to enhance the film properties, leading to low resistivity and low emissivity (the ability of a material to re-radiate

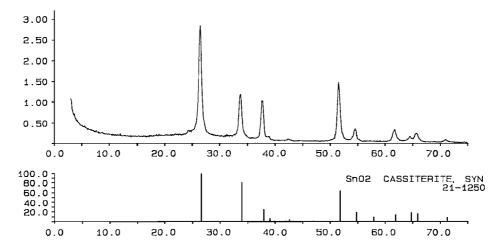


Figure 2. XRD pattern for the $F: SnO_2$ film grown from $Bu_3SnC_4F_9$ (1).

Table 5. Properties of deposited F: SnO₂ films

Precursor	1	2	3	4	5	Standarda
Thickness (nm)	379.5	367.5	616.5	200.0	433.0	300.0
Haze (%)	0.64	0.54	4.10	0.50	1.05	< 0.40
Emissivity	0.220	0.328	0.139	0.655	0.132	< 0.150
Sheet resistance $(\Omega/\Box)^b$	22	58	9	215	11	15
Resistivity ($\times 10^{-3} \Omega \text{ cm}$)	0.85	2.13	0.58	5.10	0.46	0.50
Fluorine content (at.%)	1.48	2.02	0.79	< 0.05	1.20	2.00

^a Typical measurements for a commercial fluorine-doped tin oxide film derived from separate tin and fluorine sources.⁸⁸

 $b = 25 \text{ mm}^2$

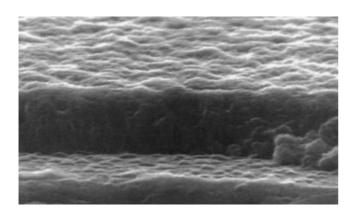


Figure 3. Scanning electron micrographs of the $F:SnO_2$ film grown from $Bu_3SnC_4F_9$ (1).

absorbed energy into a colder environment; low-*E* coatings are desirable in cold climates to prevent heat loss); good visual appearance is indicated by a low haze, which increases as the film becomes more cloudy.

All precursors produced fluorine-doped tin oxide films, establishing that the perfluoroalkyltin compounds were

capable of acting as single-source precursors. Reasonable success was achieved in trying to obtain films of approximately 300 nm thickness, the approximate dimension of commercial F:SnO₂ films, from precursors with which there were sufficient quantities to perform several runs. Owing to the small synthetic quantities of precursors 3 and 4, growth rate was not optimized and the best film derived from 3 was extremely thick (616.5 nm), whereas a very thin film (200.0 nm) was obtained from precursor 4. Data from these films can, therefore, only be taken as a guide to optimum film properties. As might be expected, low emissivity and resistivity were noted for the thick film grown from 3, but at the expense of very high haze (4.10%). In contrast, higher emissivity and resistivity were found for the thin film grown from precursor 4, although the haze was also reasonably high (0.50%) for a film of only 200.0 nm thickness. Compound 4 does not seem to be a promising precursor for F:SnO2. The properties of the film derived from precursor 3 are also not a significant improvement on the measurements given for a standard coating, given its enhanced thickness. Owing to the difficult and expensive synthesis of the precursors, and, more importantly, the production of better quality films from other precursors, it was decided not to synthesize additional material in order to perform further deposition experiments.

Films derived from 1, 2 and 5 have properties that approach those of commercial films derived from a dual-source approach, which suggests that a more detailed deposition study could lead to even more competitive properties. From the trends in data available from this initial study, it appears that increasing the number of R_f groups in the precursor diminishes the resultant film quality (1 versus 2). Moreover, data from 3 and 4 also suggest that increased fluorine content in the precursor, either from more or longer R_f groups, also has a detrimental effect on film quality. The film derived from 5, which has the best overall portfolio of properties and accrues at the highest growth rate, further implies one R_f group is sufficient and that $(C_2H_5)_nSn$ is favoured over $(C_4H_9)_nSn$, a feature that we have noted in other related precursor systems, e.g. $Et_3SnO_2CCF_3$ versus $Bu_3SnO_2CCF_3$.

The fluorine content of the films shows no discernable pattern and raises a number of issues. From compounds 1 and 2, which contain an increasing number of fluorinated groups, it can be seen that the incorporation of additional fluorine into the precursor leads to an increase in the quantity of fluorine found in the resultant tin oxide film. Furthermore, the fluorine content determined for the film grown from precursor 5 is similar to that found for 1, which is reasonable given that the compounds have an identical R₃SnC₄F₉ formulation. However, a reduced fluorine content was observed from BuSn(C₄F₉)₃ (3) with three fluorinated ligands. The reason for this is unclear, but it could suggest that an alternative or competing decomposition mechanism is operating. The extremely low fluorine content found in the film deposited from Bu₃SnC₆F₁₃ (4) was unexpected, but probably explains the extremely poor set of properties exhibited by this film (Table 5).

The incorporation of fluorine from the precursor into the SnO₂ film is inevitably related to both the structure of the precursor and its decomposition mechanism. Previous studies on the pyrolysis of fluoroorganotin compounds have established that (i) α -elimination is feasible where no other alternative is possible (decomposition of Me₃SnCF₃ to Me₃SnF and *cyclo*-C₃F₆ at 150 °C),⁵⁹ (ii) β -elimination is not as facile as might be expected (Me₃SnC₂F₅ remains 91% intact after 72 h at 200 °C; small amounts of C2F5H are detected), but it is enhanced by the presence of branchedchain fluoroalkyl groups [Me₃SnCF(CF₃)₂ decomposes to Me₃SnF and $F_2C=C(F)CF_3$ at 150 °C over a 64 h period],⁵⁷ and (iii) γ -F competes well with β -H transfer [R₃SnCH₂CH₂CF₃ decomposes to yield similar amounts of R₃SnF/cyclo-C₃H₄F₂ and $R_3SnH/H_2C=C(H)CF_3$ at 200–300 °C over 2 h; R = $C = CC_4H_9$].⁵⁵ Our structural study of Me₃SnC₄F₉ shows that approach of γ -F and δ -F to tin along reaction coordinates that favour formation of five-coordinated transition states occurs in conformers 3 and 4 respectively, with lengthening of the C-F bonds, and that transfer of fluorine from these sources is most likely. The experimental amount of conformer 3 is only 1%, but this value is very uncertain, and the calculated amount is 12% at 400 K (the temperature of the GED experiment), rising to 21% at 830 K (the temperature of the CVD experiment). Conformer 4 is calculated to account for fewer than 1% of the molecules at 400 K but 3.6% at 830 K, so it, too, could be involved in the fluorine-transfer process. In addition, if either or both of the γ - and δ -F sites are responsible for the doping, then the low fluorine content of the film produced by Bu₃SnC₆F₁₃ can plausibly be rationalized by the fact the electron-withdrawing C₂F₅ group attached to the δ -carbon in this compound effectively reduces the basicity of the fluorine atoms attached to the γ - and δ -CF₂ centres.

CONCLUSIONS

In conclusion, the films produced from the perfluoroalkyltin compounds were encouraging, and showed that it was possible to grow a fluorine-doped tin oxide film from a single-source precursor. Overall, it appears that the best arrangement for a precursor in this class consists of one containing a single fluorinated group. The length of the fluorinated chain seems to be important; for effective fluorinedoping a relatively small R_f group appears to be essential, as the fluorine incorporation diminishes as R_f changes from C_4F_9 (1) to C_6F_{13} (4). For increased volatility and, hence, a significantly shorter run time, a vast improvement is achieved by the incorporation of small R groups (Et versus Bu), although film properties appear to be unaffected by the choice. Although organotin compounds containing small R groups are more expensive and have a higher toxicity than the corresponding butyltin compounds, the CVD properties are greatly enhanced.

SUPPLEMENTARY INFORMATION

A complete set of calculated geometrical parameters for $Me_3SnC_4F_9$ (distances in picometres) from the *ab initio* MO theory study (Table 2), interatomic distances (r_a/pm) and amplitudes of vibration (u/pm) for the restrained GED structure of $Me_3SnC_4F_9$ (Table 6), least-squares correlation matrix (×100) for $Me_3SnC_4F_9$ (Table 7), experimental and difference (experimental minus theoretical) radial-distribution curves, P(r)/r, for $Me_3SnC_4F_9$ (Figure 4) and experimental and final weighted difference (experimental minus theoretical) molecular-scattering intensities for $Me_3SnC_4F_9$ (Figure 5).

Acknowledgements

We thank the Engineering and Physical Sciences Research Council for research studentships (B. F. J. and J. E. S.) and for support for the electron diffraction service (grant GR/R17768). We also thank Dr V. Typke of the University of Ulm for the variable-array version of the ASYM40 program, Dr S. L. Hinchley (Edinburgh) for assistance and Pilkington plc for financial support and help with the film analysis.

REFERENCES

- 1. Fillard JP, Manifacier JC. Jpn. J. Appl. Phys. 1970; 9: 1012.
- 2. Tatsuyama C, Ichimura S. Jpn. J. Appl. Phys. 1976; 15: 843.

J. E. Stanley et al.

- 3. Hass G, Heaney JB, Toft AR. Appl. Opt. 1979; 18: 1488.
- 4. Brinker DJ, Wang EY, Wadlin WH, Legge RN. J. Electrochem. Soc. 1981; 128: 1968.
- 5. Chopra KL, Major S, Pandya DK. Thin Solid Films 1983; 102: 1.
- 6. Das SK, Morris GC. J. Appl. Phys. 1993; 73: 782.
- Ferrere S, Zaban A, Gregg BA. J. Phys. B: At. Mol. Opt. Phys. 1997; **101**: 4490.
- 8. Ford WE, Wessls JM, Rodgers MAJ. J. Phys. Chem. B 1997; 101: 7435.
- 9. Laverty SJ, Feng H, Maguire P. J. Electrochem. Soc. 1997; 144: 2165.
- 10. Bélanger D, Dodelet JP, Lombos BA, Dickson JI. J. Electrochem. Soc. 1985; 132: 1398.
- 11. Upadhyay JP, Vishwakarma SR, Prasad HC. Thin Solid Films 1989; 169: 195
- 12. Vishwakarma SR, Upadhyay JP, Prasad HC. Thin Solid Films 1989; 176: 99.
- 13. Haneko H, Miyake K. J. Appl. Phys. 1992; 53: 3629.
- 14. Lin Y-J, Wu C-J. Surf. Coat. Technol. 1996; 88: 239.
- 15. Proscia J, Gordon RG. Thin Solid Films 1992; 214: 175.
- 16. Ishida T, Tabata O, Park JW, Shin SH, Magara H, Tamura S, Mochizuki S, Mihara T. Thin Solid Films 1996; 281-282: 228.
- 17. Acosta DR, Zironi EP, Montoya E, Estrada W. Thin Solid Films 1996; 288: 1.
- 18. Chatelon JP, Terrier C, Bernstein E, Berjoan R, Roger JA. Thin Solid Films 1994; 247: 162.
- 19. Park S-S, Mackenzie JD. Thin Solid Films 1995; 258: 268.
- 20. Racheva TM, Critchlow GW. Thin Solid Films 1997; 292: 299.
- 21. Ray SC, Karanjai MK, DasGupta D. Surf. Coat. Technol. 1998; 102:
- 22. Gamard A, Jousseaume B, Toupance T, Campet G. Inorg. Chem. 1999; 38: 4671.
- 23. Gamard A, Babot O, Jousseaume B, Rascle M-C, Toupance T, Campet G. Chem. Mater. 2000; 12: 3419.
- 24. Minami T, Nanto H, Takata S. Jpn. J. Appl. Phys. 1988; 27: L287.
- 25. Geoffroy C, Campet G, Menil F, Portier J, Salardenne J, Couturier G. Active Passive Elec. Comput. 1991; 14: 111.
- 26. Karlsson T, Roos A, Ribbing C-G. Sol. Energy Mater. 1985; 11: 469.
- 27. Yagi I, Ikeda E, Kuniya Y. J. Mater. Res. 1994; 9: 663.
- 28. Gordillo G, Moreno LC, de la Cruz W, Teheran P. Thin Solid Films 1994; **252**: 61.
- 29. Laurent J-M, Smith A, Smith DS, Bonnet J-P, Clemente RR. Thin Solid Films 1997; 292: 145.
- 30. Shanthi S, Subramanian C, Ramasamy P. Mater. Sci. Eng. B 1999; 57: 127.
- 31. Karlsson B, Valkonen E, Karlsson T, Ribbing C-G. Thin Solid Films 1981; 86: 91.
- 32. Granqvist CG. Thin Solid Films 1990; 193: 730.
- 33. Nikodem RB. J. Vac. Sci. Technol. A 1992; 10: 1884.
- 34. Ghandhi SK, Siviy R, Borrego JM. Appl. Phys. Lett. 1979; 34: 833.
- 35. Borman CG, Gordon RG. J. Electrochem. Soc. 1989; 136: 3820.
- 36. Wan CF, McGrath RD, Keenan WF, Frank SN. J. Electrochem. Soc. 1989; 136: 1459.
- 37. Ghoshtagore RN. J. Electrochem. Soc. 1978; 125: 110.
- 38. Kojima M, Kato H, Imai A, Yoshida A. J. Appl. Phys. 1988; 64:
- 39. Yoon KH, Song JS. Thin Solid Films 1993; 224: 203.
- 40. Smith A, Laurent J-M, Smith DS, Bonnet J-P, Clemente RR. Thin Solid Films 1995; 266: 20.
- 41. Yusta FJ, Hitchman ML, Shamlian SH. J. Mater. Chem. 1997; 7: 1421.
- 42. Atagi LM, Hoffman DM, Liu J-R, Zheng Z, Chu W-K. Chem. Mater. 1994; 6: 360.
- 43. Tan C, Xia Y, Chen Y, Li S, Liu J, Liu X, Xu B. J. Appl. Phys. 1983; 73: 4266.
- 44. Ma HL, Zhang DH, Win SZ, Li SY, Chen YP. Sol. Energy Mater. 1996; **40**: 371.
- 45. Yoon KH, Song JS. Sol. Energy Mater. 1993; 28: 317.

- 46. Saxena AK, Thangaraj R, Singh SP, Agnihotri OP. Bull. Mater. Sci. 1986; 8: 315.
- 47. Athey PR, Urban III FK, Holloway PH. J. Vac. Sci. Technol. B 1996; 14: 3436.
- 48. Clark HC, O'Brien RJ, Trotter J. J. Chem. Soc. C 1964; 2332.
- 49. Tudela D, Gutiérrez-Puebla E, Monge A. J. Chem. Soc. Dalton Trans. 1992; 1069.
- 50. Boegeat D, Jousseaume B, Toupance T, Campet G, Fournès L. Inorg. Chem. 2000; 39: 3924.
- 51. Suh S, Hoffman DM, Atagi LM, Smith DS, Liu J-R, Chu W-K. Chem. Mater. 1997; 9: 730.
- 52. Suh S, Hoffman DM. Inorg. Chem. 1996; 35: 6164.
- 53. Maruyama T, Tabata K. J. Appl. Phys. 1990; 68: 4282.
- 54. Suh S, Zhang Z, Chu W-K, Hoffman DM. Thin Solid Films 1999; 345: 240.
- 55. Franc C, Jousseaume B, Linker M, Toupance T. Chem. Mater. 2000; 12: 3100.
- 56. Boutet S, Gamard A, Jousseaume B, Toupance T, Campet G, Cachet H. Main Group Met. Chem. 2002; 25: 59.
- 57. Cullen WR, Sams JR, Waldman MC. Inorg. Chem. 1970; 9: 1682.
- 58. Keasz HD, Phillips JR, Stone FGA. J. Am. Chem. Soc. 1960; 82: 6228.
- 59. Clark HC, Willis CJ. J. Am. Chem. Soc. 1960; 82: 1888.
- 60. Uno H, Shiraishi Y, Suzuki H. Bull. Chem. Soc. Jpn. 1989; 62: 2636.
- 61. Stone FGA, Treichel PM. Chem. Ind. (London) 1960; 837.
- 62. Eujen R, Jahn N, Thurmann U. J. Organometal. Chem. 1994; 465: 153.
- 63. Seyferth D, Richter F. J. Organometal. Chem. 1995; 499: 131.
- 64. Kawakami K, Kuivila HG. J. Org. Chem. 1969; 34: 1502.
- 65. Burdon J, Coe PL, Haslock IB, Powell RL. J. Chem. Soc. Chem. Commun. 1996; 49.
- 66. Krause LJ, Morrison JA. J. Chem. Soc. Chem. Commun. 1980; 671.
- 67. Kitazume T, Ishikawa N. Chem. Lett. 1981; 1337.
- 68. Molloy KC, Purcell TG, Quill K, Nowell I. J. Organometal. Chem. 1984; 267: 237.
- 69. Edwards DA, Harker RM, Mahon MF, Molloy KC. J. Mater. Chem. 1999: 9: 1771.
- 70. Kaelble EF. Handbook of X-Rays. McGraw-Hill: New York, 1967; chapter 17.
- 71. Bass M. Handbook of Optics, 2nd edition. McGraw Hill: New York,
- 72. Frisch MJ, Trucks GW, Schlegel HB, Gill PMW, Johnson BG, Robb MA, Cheesman JR, Keith TA, Petersson GA, Montgomery JA, Raghavachari K, Al-Laham MA, Zakrzewski VG, Ortiz IV, Foresman JB, Cioslowski J, Stefanov BB, Nanayakkara A, Challacombe M, Peng CY, Ayala PY, Chen W, Wong MW, Andres JL, Replogle ES, Gomperts R, Martin RL, Fox DJ, Binkley JS, Defrees DJ, Baker J, Stewart JP, Head-Gordon M, Gonzalez C, Pople JA. Gaussian 94 (Revision C.2). Gaussian Inc., Pittsburgh, PA, 1995.
- 73. Dunning TH, Hay PJ. In Modern Theoretical Chemistry, Schaefer I HF (ed.). Plenum: New York; 1.
- 74. Hay PJ, Wadt WR. J. Chem. Phys. 1985; 82: 270.
- 75. Wadt WR, Hay PJ. J. Chem. Phys. 1985; 82: 284.
- 76. Hay PJ, Wadt WR. J. Chem. Phys. 1985; 82: 299.
- 77. Hedberg L, Mills IM. J. Mol. Spectrosc. 1993; 160: 117.
- 78. Scott AP, Radom L. J. Phys. Chem. 1996; 100: 16502.
- 79. Huntley CM, Laurenson GS, Rankin DWH. J. Chem. Soc. Dalton Trans. 1980; 954.
- 80. Hinchley SL, Robertson HE, Borisenko KB, Turne AR, Johnson BF, Rankin DWA, Ahmadlan M, Jones JN, Lowley AH. J. Chem. Soc. Dalton Trans. 2004; 2469.
- 81. Ross AW, Fink M, Hilderbrand R. In International Tables for Crystallography, Wilson AJC (ed.). Kluwer Academic Publishers: Dordrecht; 245.



- Blake AJ, Brain PT, McNab H, Miller J, Morrison CA, Parsons S, Rankin DWH, Robertson HE, Smart BA. J. Chem. Phys. 1996; 100: 12 280.
- 83. Mitzel NW, Smart BA, Blake AJ, Robertson HE, Rankin DWH. J. Phys. Chem. 1996; 100: 9339.
- 84. Mitchell TN. J. Organometal. Chem. 1973; 59: 189.
- 85. Nádvorník M, Holecek J, Handlir K, Lycka A. J. Organometal. Chem. 1984; 275: 43.
- 86. Parish RV, Platt RH. J. Chem. Soc. A 1969; 2145.
- 87. Agashe C, Marathe BR. J. Phys. D: Appl. Phys. 1993; 26: 2049.

- 88. Soubeyrand MJ, Halliwell AC. US Patent 5 698 262, 1997.
- 89. Fantini M, Torriani I. Thin Solid Films 1986; 138: 255.
- 90. Kwon CW, Campet G, Portier J, Poquet A, Fournès L, Labrugère C, Jousseaume B, Toupance T, Choy JH Subramanian MA. *Int. J. Inorg. Mater.* 2001; **3**: 211.
- 91. Stanley JE, Molloy KC, Mahon MF, Rankin DWH, Robertson HE, Johnston BF. Atmospheric pressure chemical vapour deposition of fluorine-doped tin(IV) oxide from fluoroalkyltin precursors. 2005; In press.